

Cement paste rheology and microstructure

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open questions

- what is the origin of cohesion of cement paste?
- what are the phenomena which cause setting (stiffening)?
 - ✓ precipitation of hydrates?
 - ✓ growth of CSH nuclei?
 - ✓ aging?
- what causes the differences in paste rheology with different admixtures?
- how can we control
 - ✓ hydration rates?
 - ✓ microstructure?
 - ✓ mechanical properties?

origin of cohesion

C3S and C-S-H in suspension: at high pH ≈ 13 the silanol surface groups will hydrolyze:



the surfaces will be negative with high charge density:

$$\sigma \approx 0,8 \text{ C/m}^2 \text{ at pH } 14$$

$$\sigma \approx 0,2 \div 0,4 \text{ C/m}^2 \text{ at pH } 12$$

why can surfaces with like charges attract?

Counterions at Highly Charged Interfaces: From One Plate to Like-Charge Attraction

Ladislav Šamaj* and Emmanuel Trizac

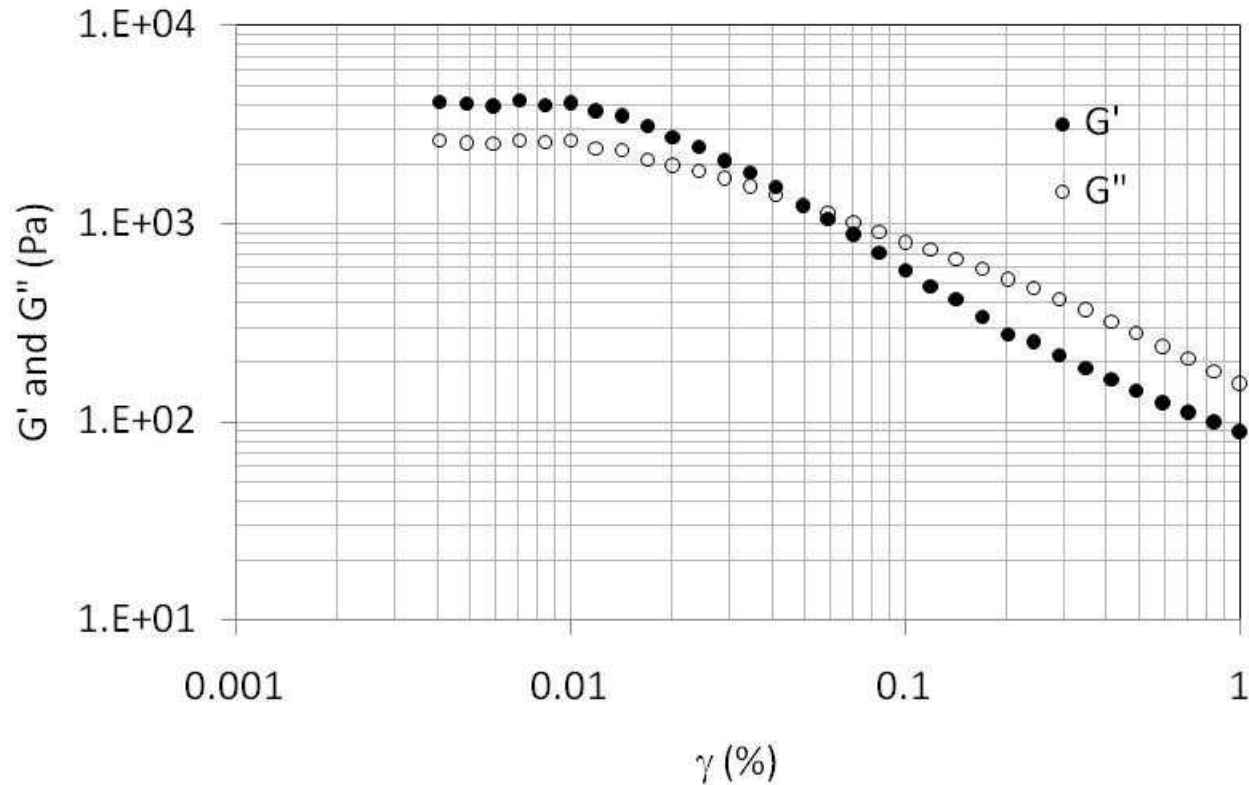
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The behavior of charged particles in the vicinity of charged interfaces is a central yet elusive problem in the equilibrium statistical mechanics of Coulomb fluids, including colloidal science. A landmark in the field was the realization in the 1980s that **similarly charged surfaces may attract** each other under strong enough Coulombic couplings, which can be realized in practice increasing the valency of the counterions involved [1]. Notorious illustrations of this like-charge attraction are the formation of DNA condensates [2] or **aggregates of colloidal particles** [3].

correlation forces

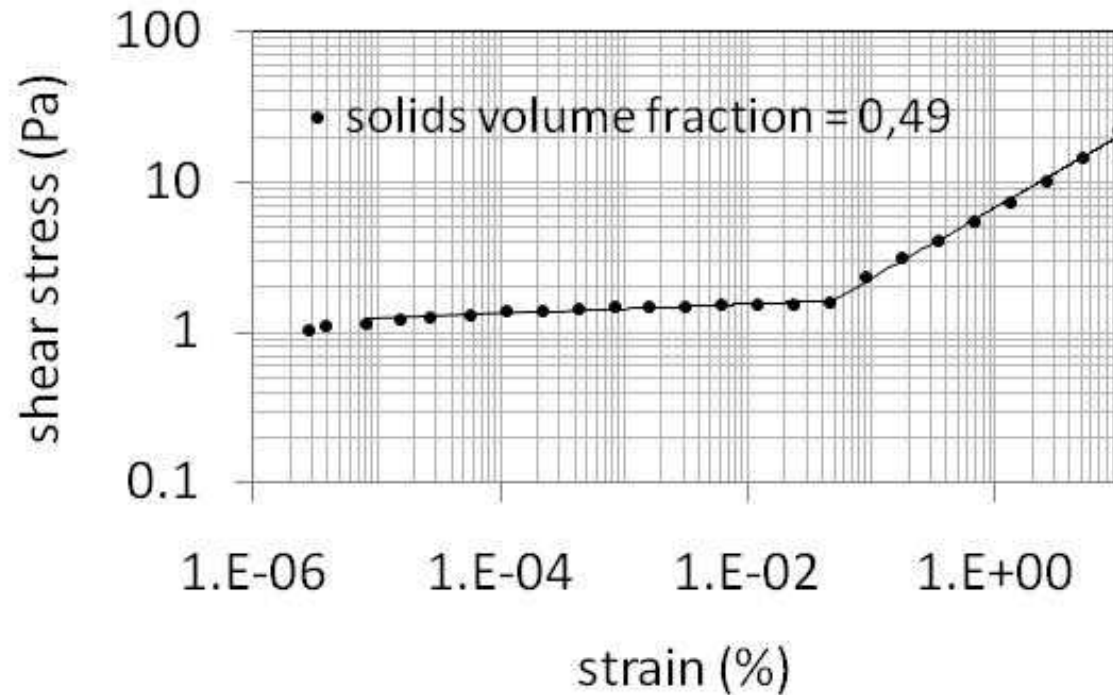
experimentally cement paste is an attractive colloid



$w/c = 0,5$
 $\Phi = 0,39$

just after mixing: the paste is elastic
and yields for large enough strains

experimentally cement paste is an attractive colloid



yield stress can be measured
also with a continuous
deformation: yield stress is
equal, yield strain is larger

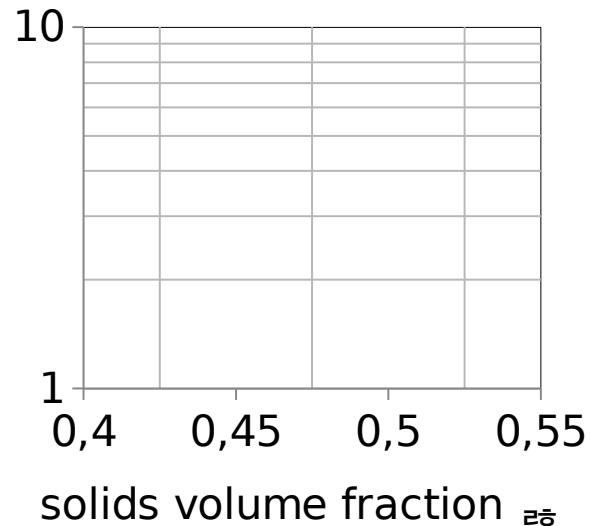
flow start-up is dilatant

just after mixing: the paste is elastic
and yields for large enough strains

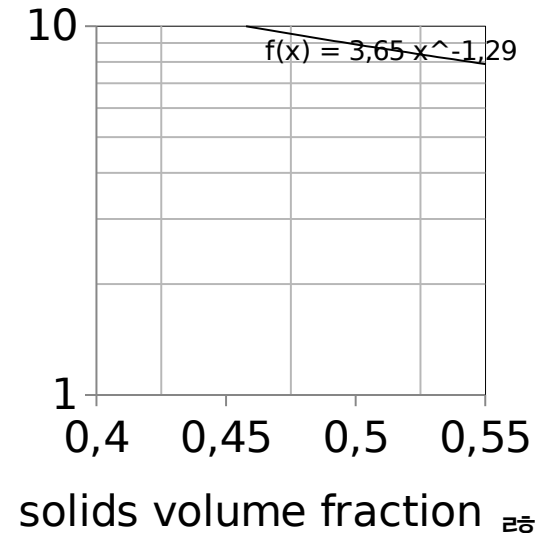
yield stress and yield strain exhibit exponential scaling with particle concentration

$$f(x) = 0,17 x^{-1,25}$$

yield strain $\epsilon_{\lambda 0}$ (Pa)

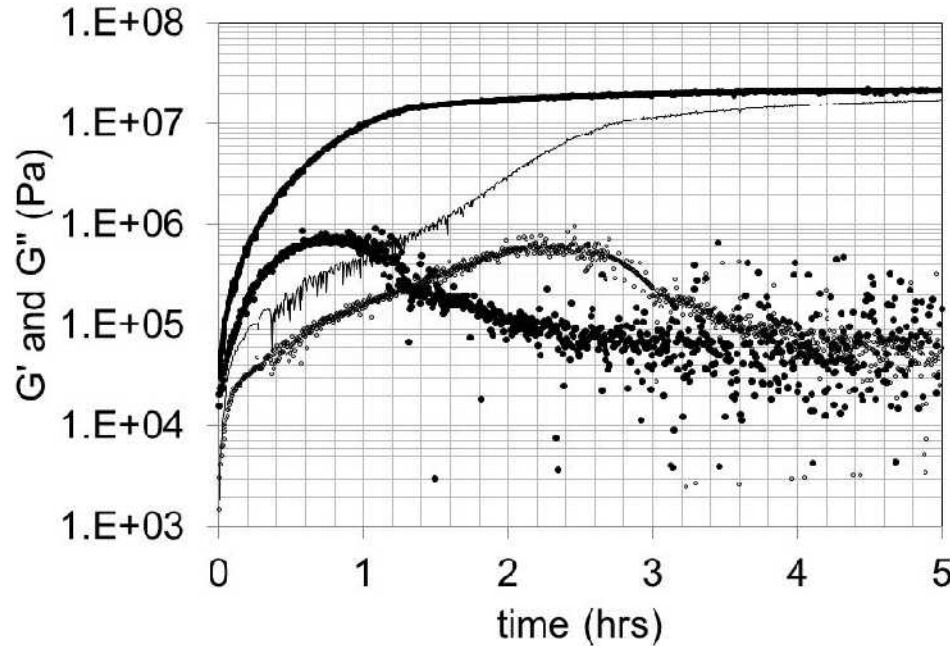


yield stress σ_0 (Pa)



the scaling exponent is a function of microstructure

cement paste undergoes a fast stiffening

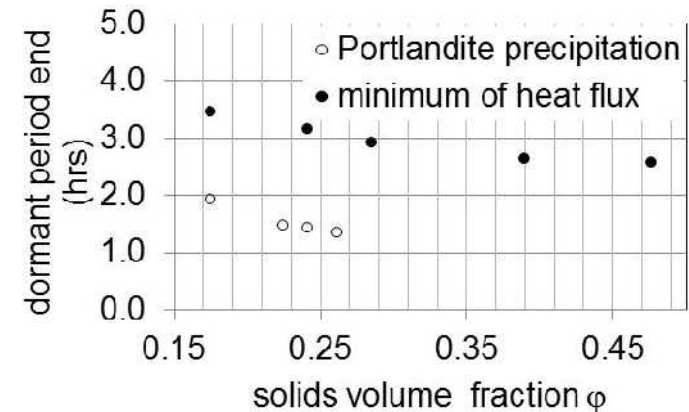


— G' w/c 0,35 · G'' w/c 0,35

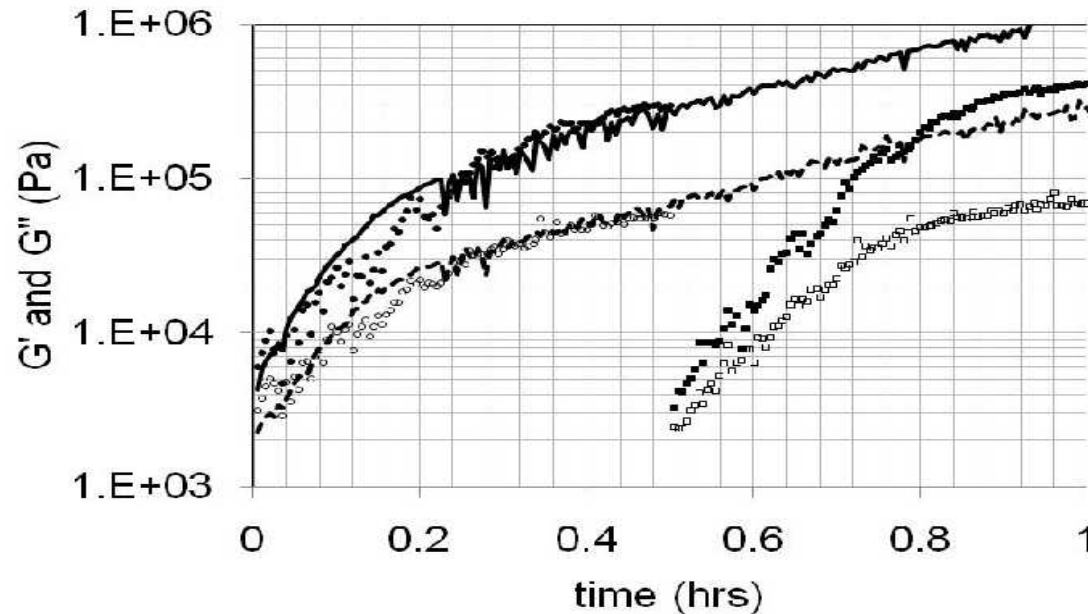
— G' w/c 0,50 · G'' w/c 0,50

during the dormant period

at Portlandite precipitation there is an upwards inflection



stiffening is reversible before Portlandite precipitation



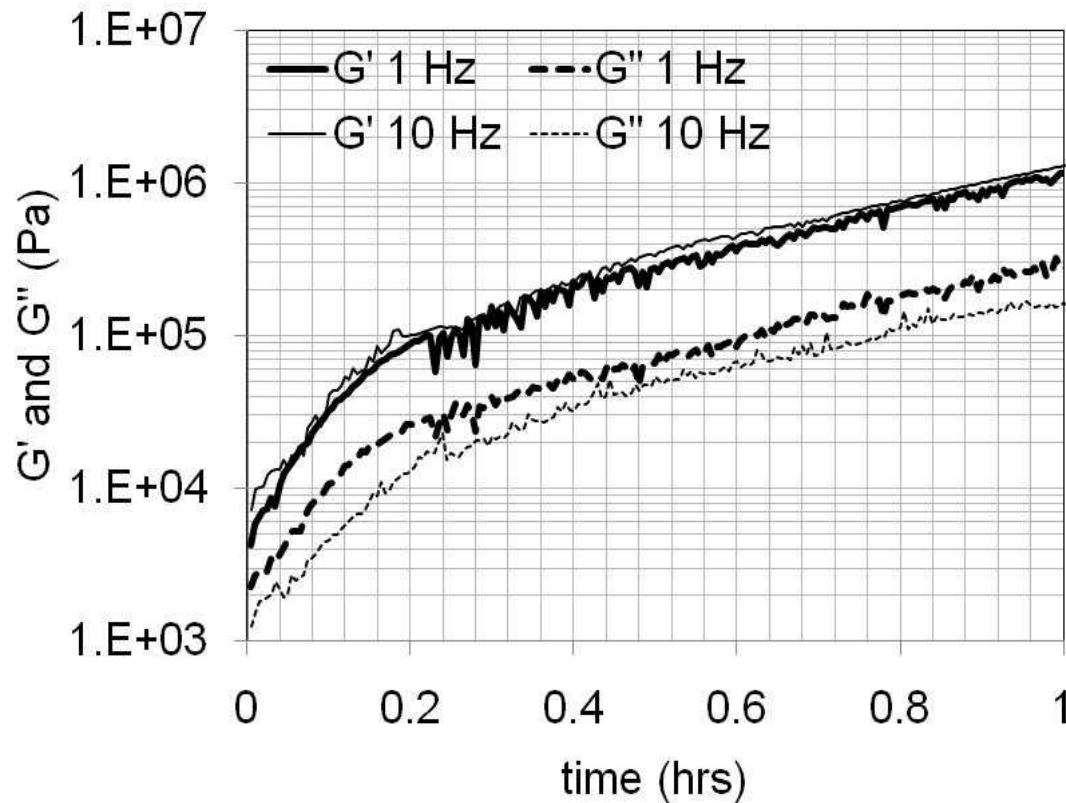
- G' w/c 0,40
- G' first rise
- G' second rise
- G'' w/c 0,40
- G'' first rise
- ◻ G'' second rise

“shearing interrupts aging”

or

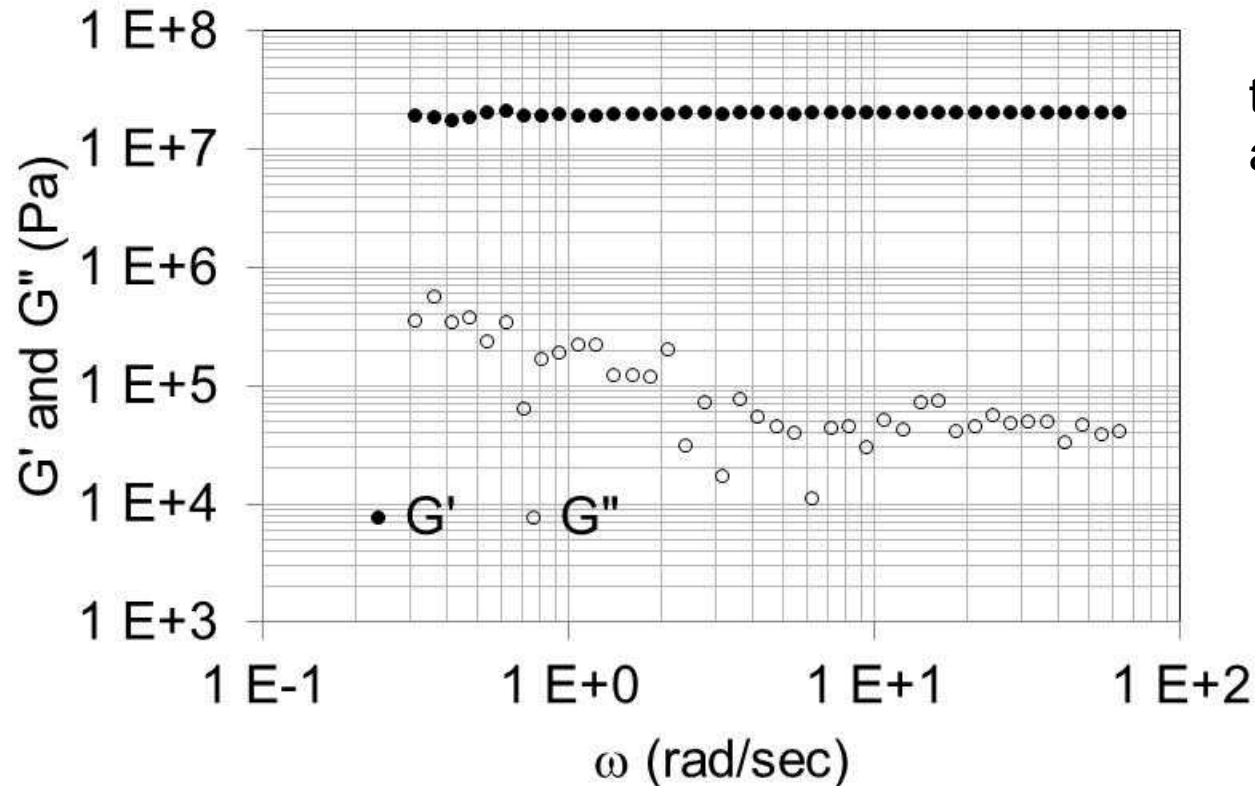
“shearing resets the distribution of heterogeneities”

the loss modulus decreases on increasing frequency

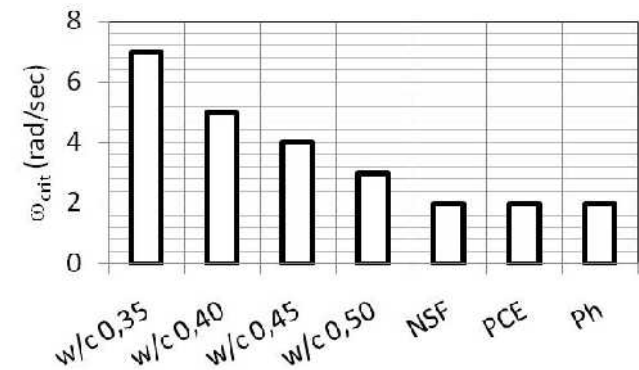


typical of an attractive gel: energy dissipation is due to particle extraction from its local position into the fluid

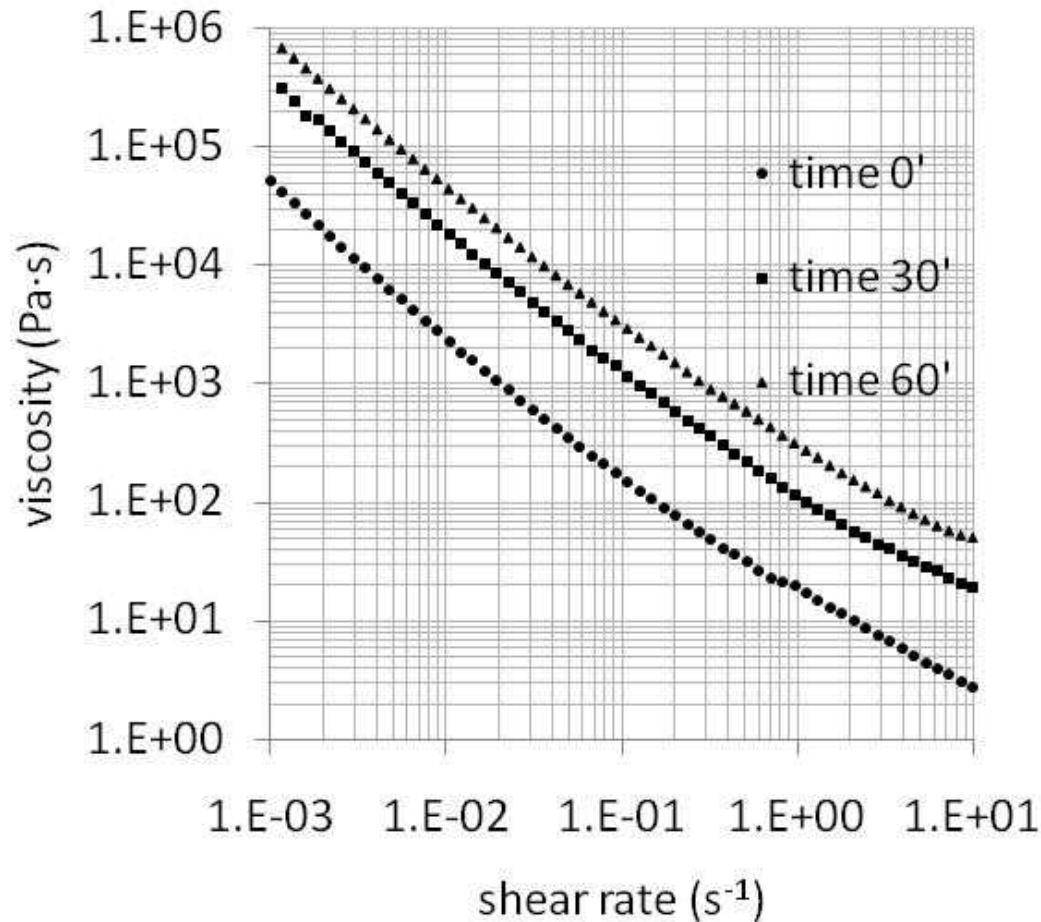
after Portlandite precipitation $G'(\omega)$ is constant and $G''(\omega)$ decreases up to ω_{crit} function of concentration



the elastic properties of paste are a function of collective motion



in flow cement paste is shear thinning at low shear rates and shear thickening at high shear rates



w/c 0,48
 $\Phi = 0,40$

viscosity increases over time, as a function of the shear imposed before measuring

shear thinning can be explained by:

- stress distribution and the formation of stress bearing structures

(Silbert, Farr, Melrose, Ball)

4786 J. Chem. Phys., Vol. 111, No. 10, 8 September 1999

Silbert *et al.*

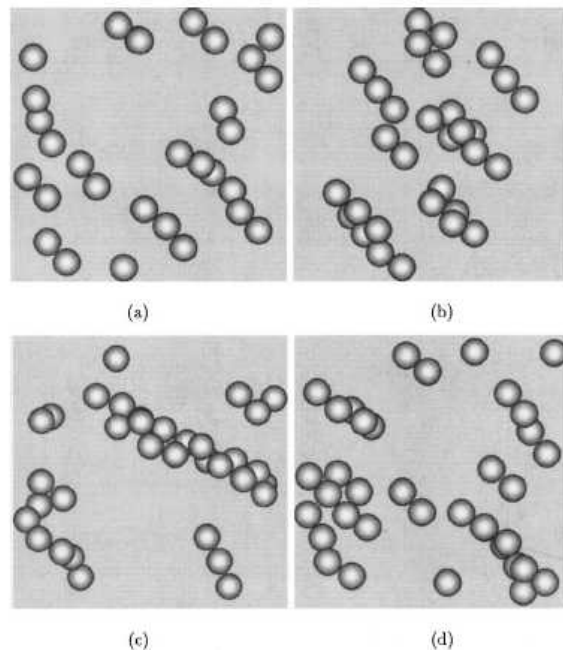


FIG. 8. A qualitative measure of the kinetics of stress bearing clusters at $Pe=1.0$, for a 700-particle system, where only those particles whose bonds carry a stress greater than $\sigma_+^c=0.204$, are shown. Strain values are (a) 106.000, (b) 106.013, (c) 107.001, and (d) 289.001.

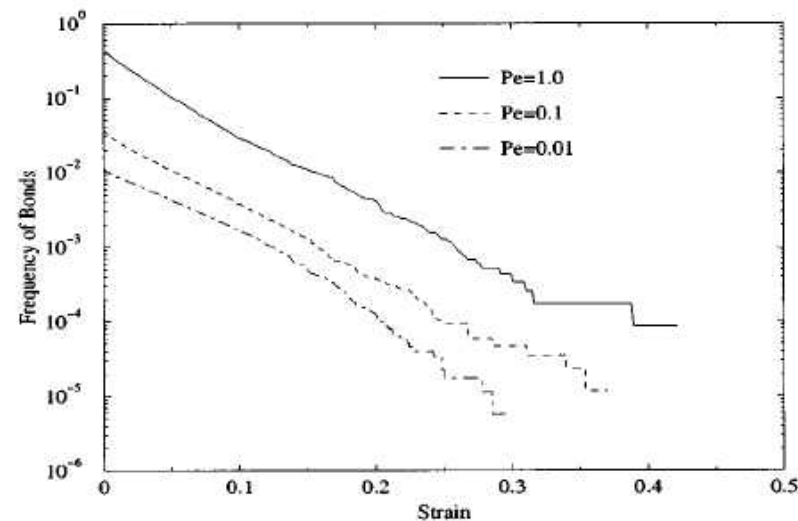


FIG. 9. Distribution of high-stress cluster bond lifetimes at $Pe=0.01$, 0.1 , and 1.0 (in units of strain γ). Cluster bonds are defined as those bonds experiencing a stress higher than the average bond stress by a factor of ~ 25 .

increasing shear rate increases cluster persistence

shear thinning can be explained by:

- agglomerate formation and disruption induced by the hydrodynamic field

(Snabre and Mills; Potanin; Morbidelli et al.)

P. Snabre ^(1,*) and P. Mills ⁽²⁾ JOURNAL DE PHYSIQUE III *September 1996*,

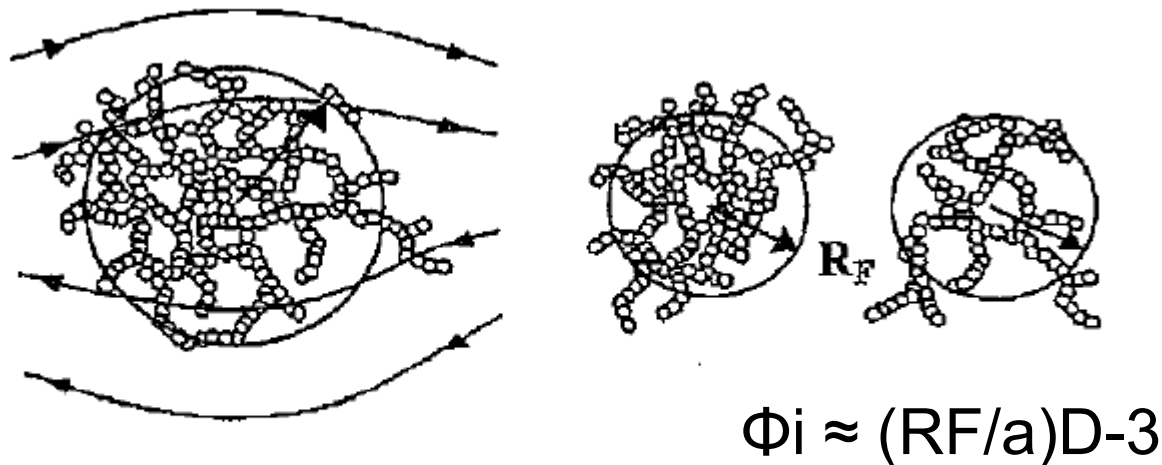
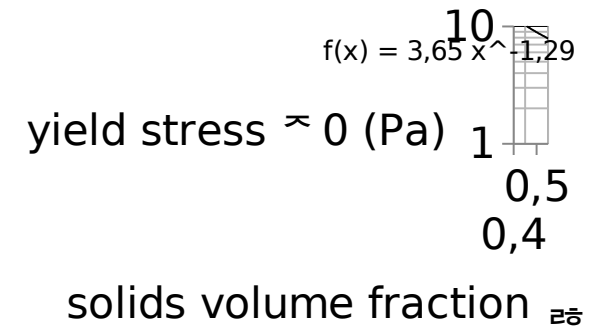


Fig. 3. — Shear break-up of a fractal cluster into approximately equal parts.

- ⇒ describing the agglomerates does not imply fractal scaling or growth: only density variation with size
- ⇒ fractal dimension is high $D \cong 2,8$: aggregates are almost compact, but not homogeneous
- ⇒ this approach delivers the fractal dimension of the agglomerates, related to porosity, and the agglomerate breakup stress, related to interaction energy



- modeling of the agglomerate breakup region gives agglomerate strength, related to interaction energy

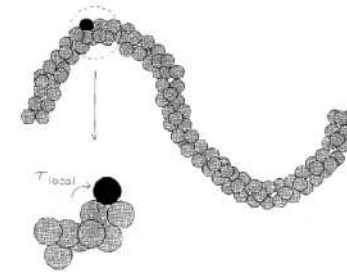
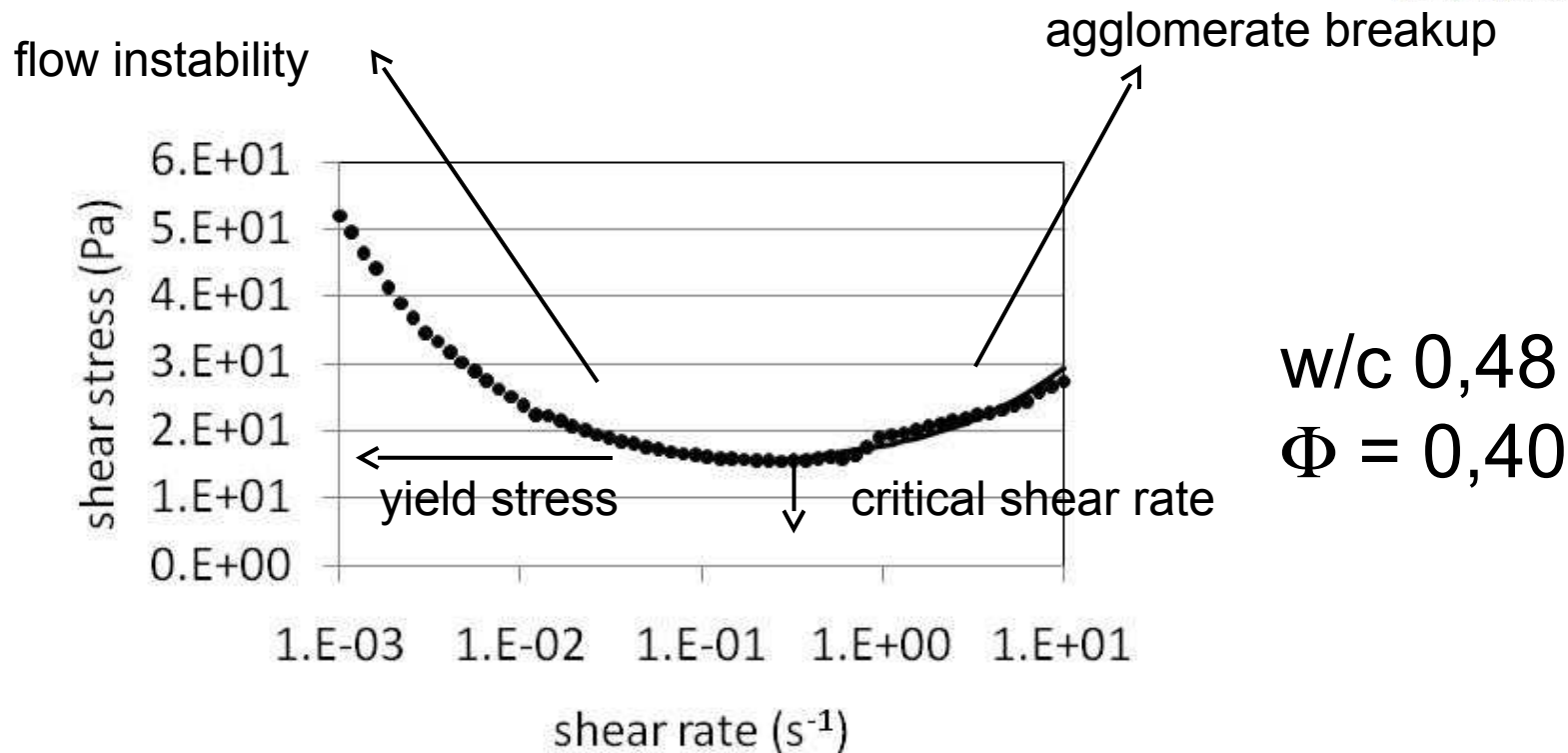
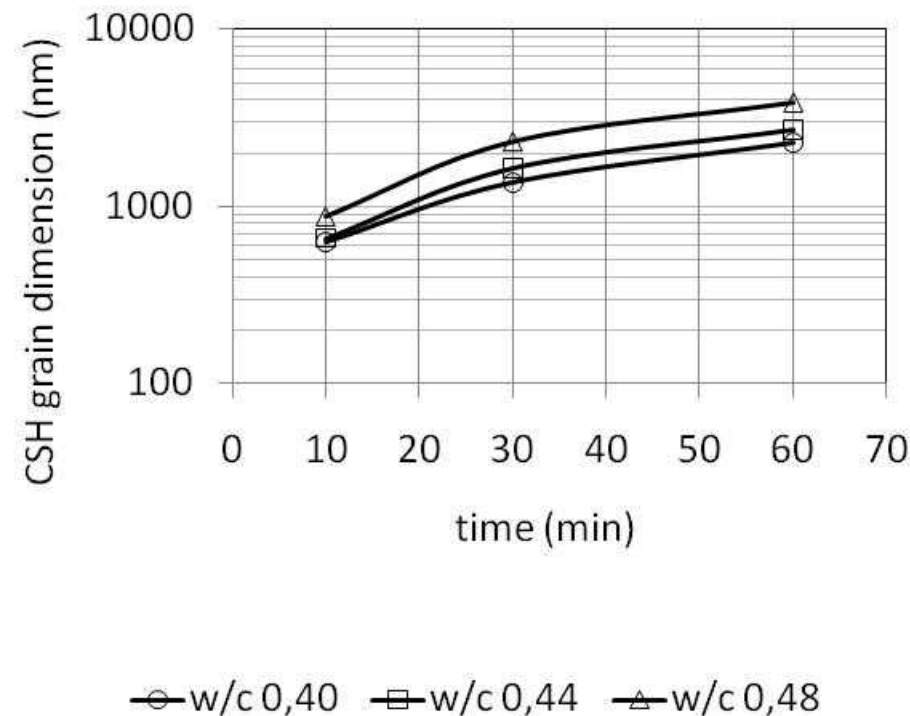


FIG. 1. The multiply connected chain.



- in the hypothesis that the connecting surfaces of the elastic chain are C-S-H nuclei it is possible to deduce dimensions



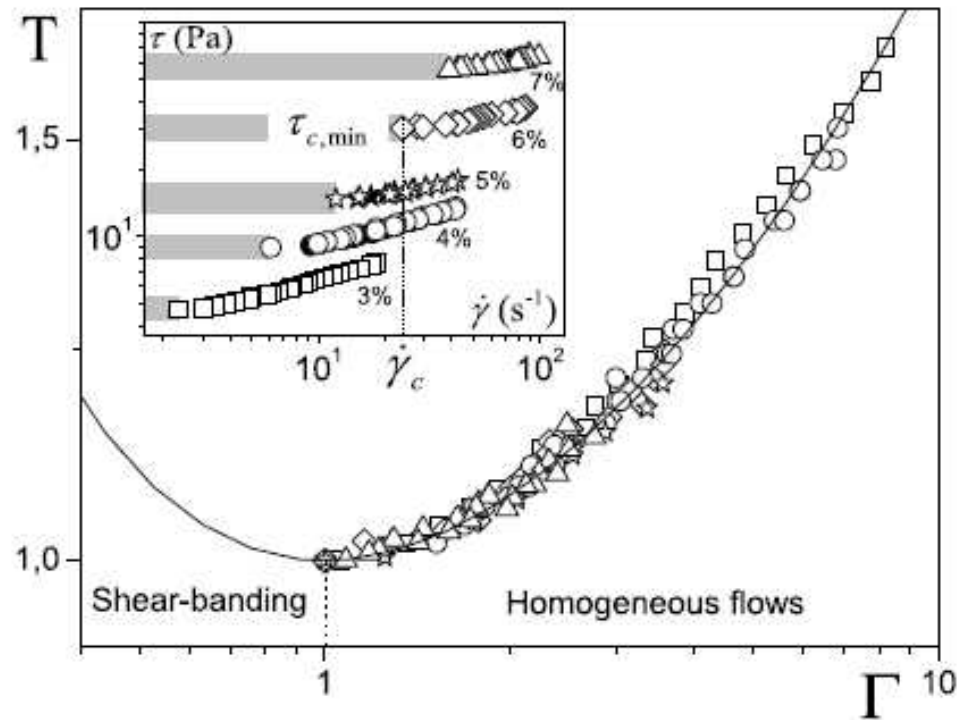
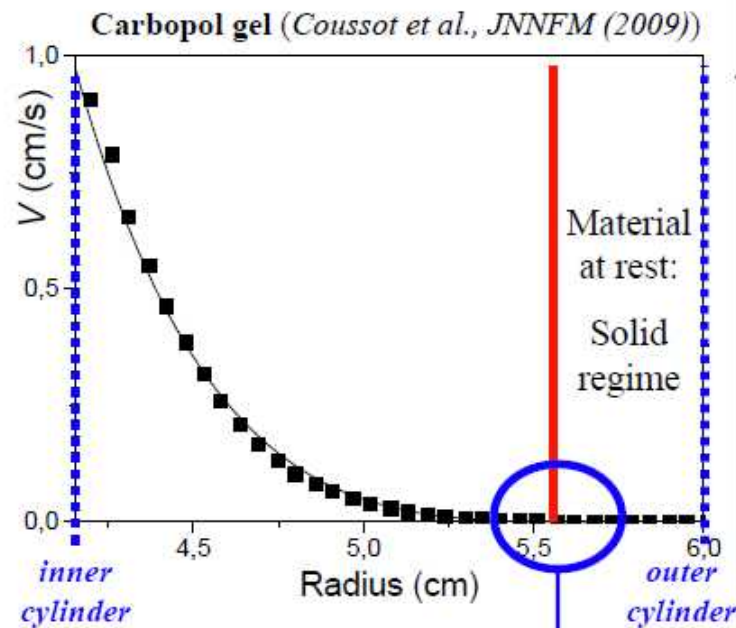


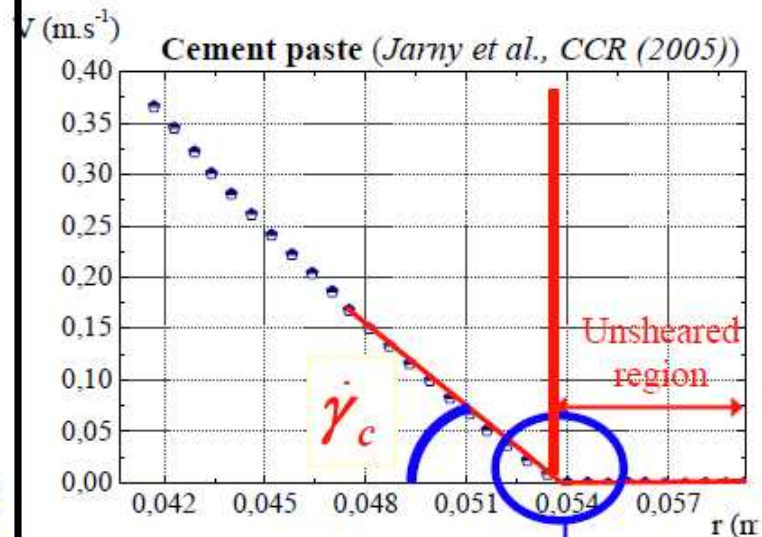
Fig. 5. Flow curves for bentonite suspensions at different solid fractions in terms of dimensionless shear rate and shear stress (see text). The continuous line is the model fitted to data with $n = 0.36$. Inset: effective flow curves (filled and empty symbols correspond to different rotation velocities of the inner cylinder); no steady flow in the grey regions.

Shear localization / Shear banding in a heterogeneous stress field



$$\tau \rightarrow \tau_Y \text{ and } \dot{\gamma} \rightarrow 0$$

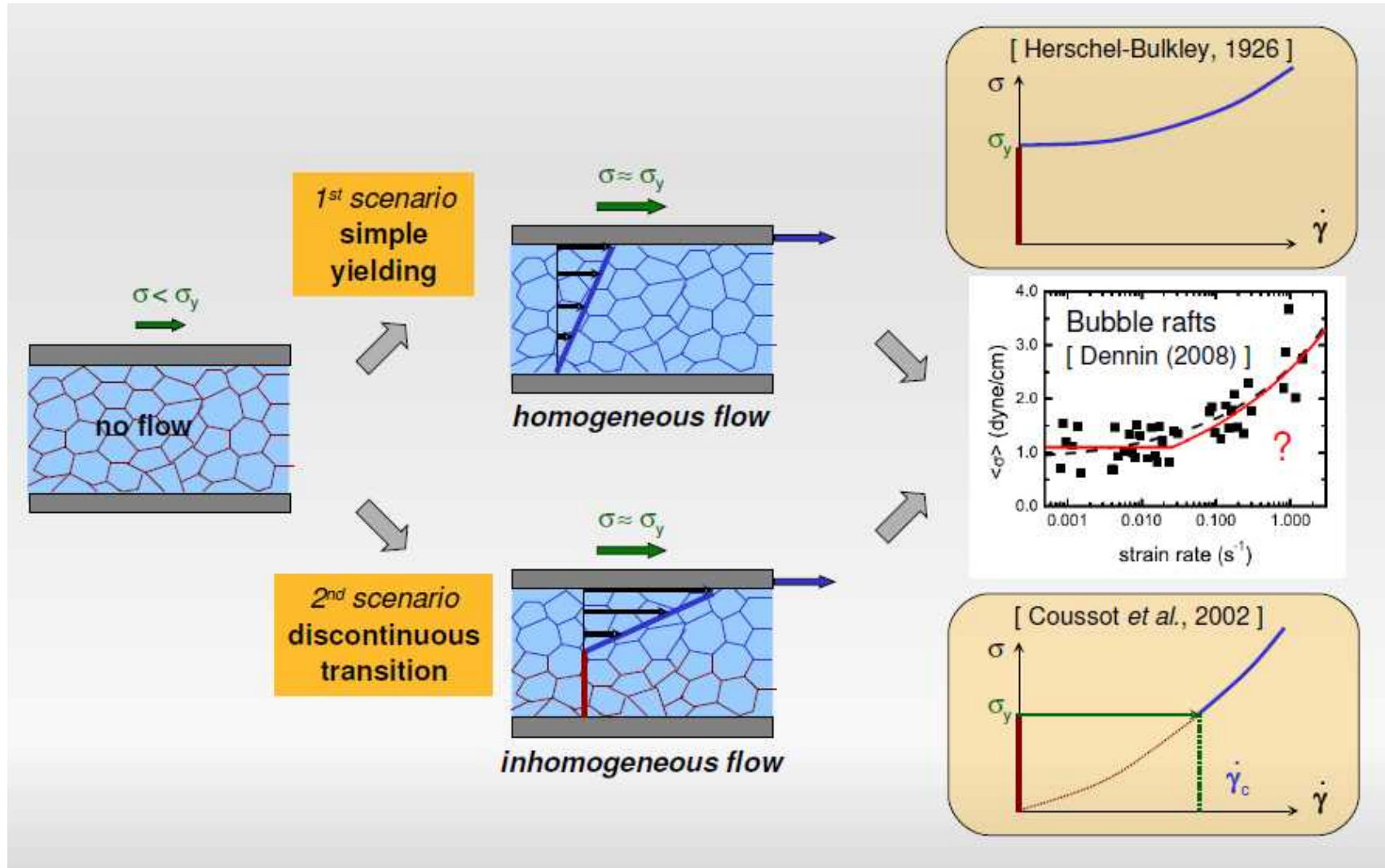
Mostly « **simple** » yield stress fluids



$$\tau \rightarrow \tau_Y \text{ and } \dot{\gamma} \rightarrow \dot{\gamma}_c$$

Mostly **thixotropic** yield stress fluids
⇐ competition between **structuration**
and **shear-induced destructuration**

flow instability and shear banding



admixtures to improve cohesivity and robustness of SCC

Component	kg/m ³
cement CEM II/A-M (LL-S) 42,5 R	429
limestone filler	90
sand 0/2 mm	860
aggregate 4/8 mm	264
aggregate 8/16 mm	605
water	189

paste: 362 l/m³

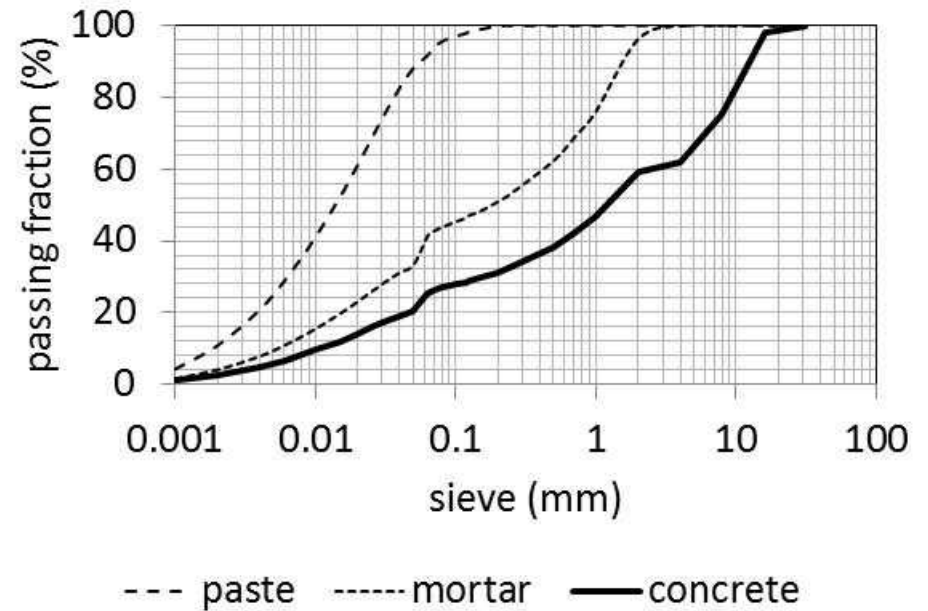
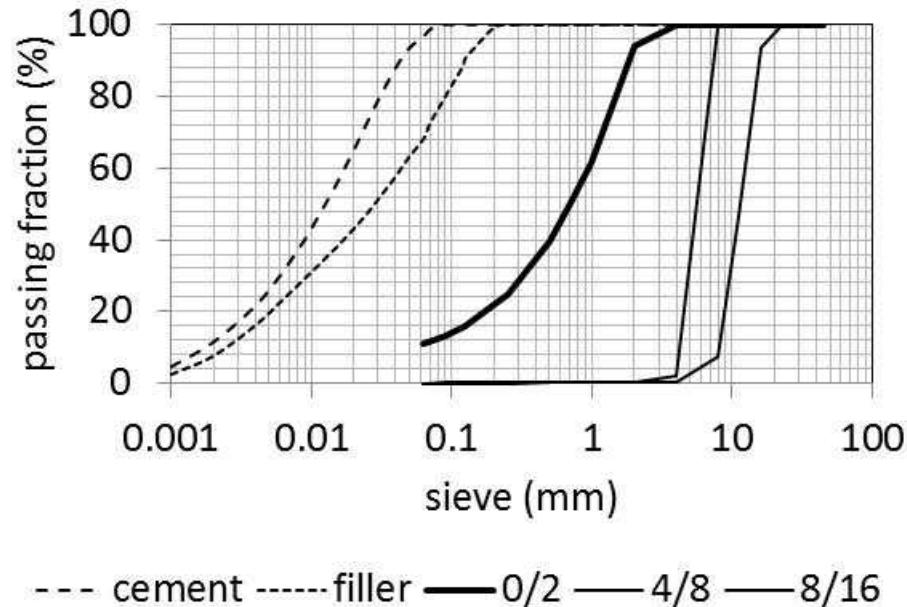
water/fines by weight: 0,36

water/fines by volume: 1,09

paste solids volume fraction: 0,48

sand/all aggregates: 0,5

admixtures to improve cohesivity and robustness of SCC



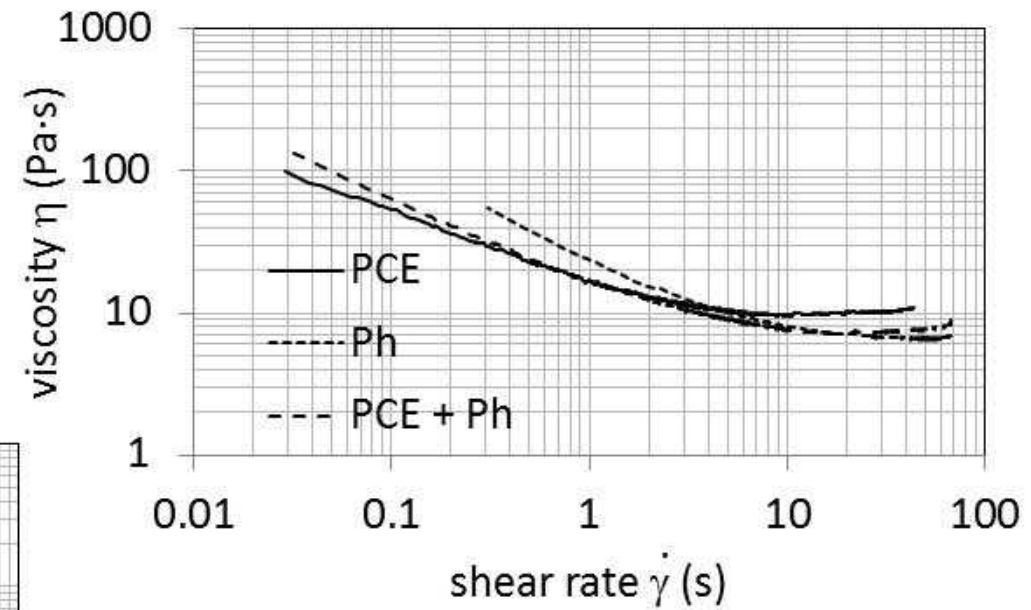
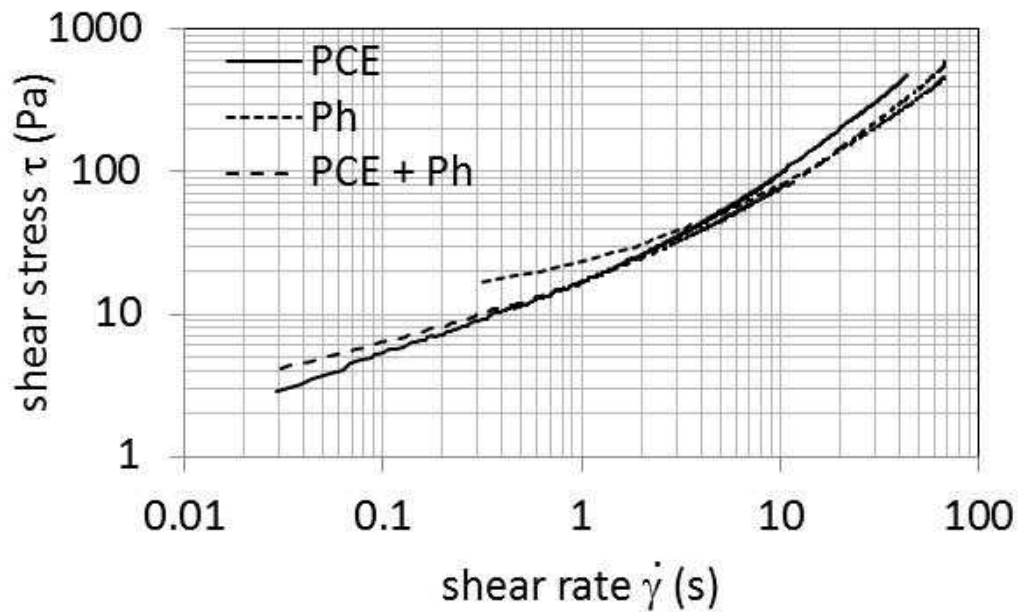
admixtures to improve cohesivity and robustness of SCC

property	PCE	Ph	PCE + Ph
slump flow (mm)	700	700	715
T ₅₀₀ (s)	3,3	2,9	2,6
V-funnel flow time (s)	8,5	6,3	6,5
L-box passing ratio	0,73	0,75	0,80

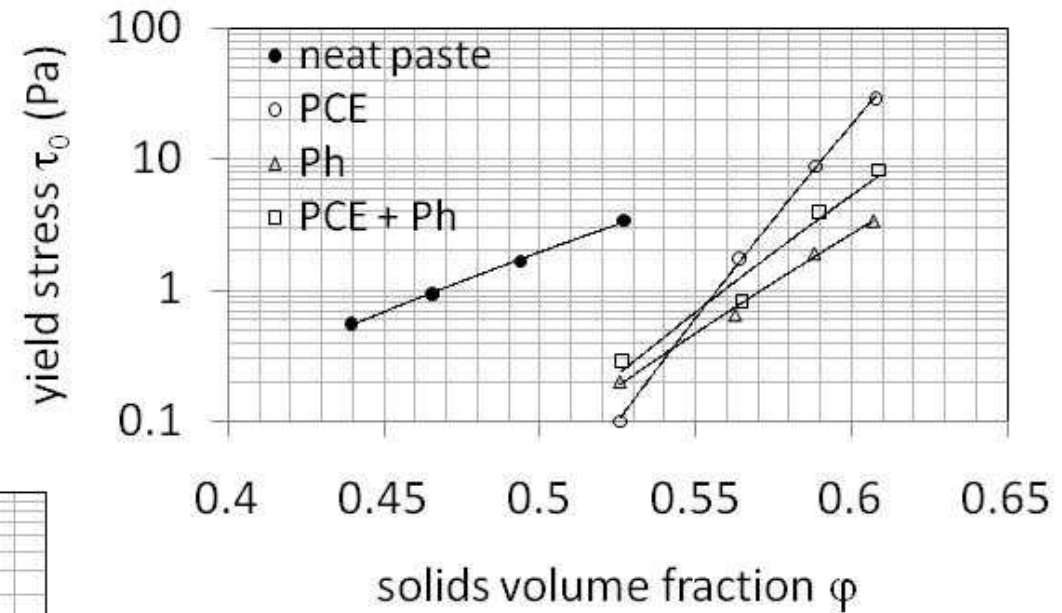
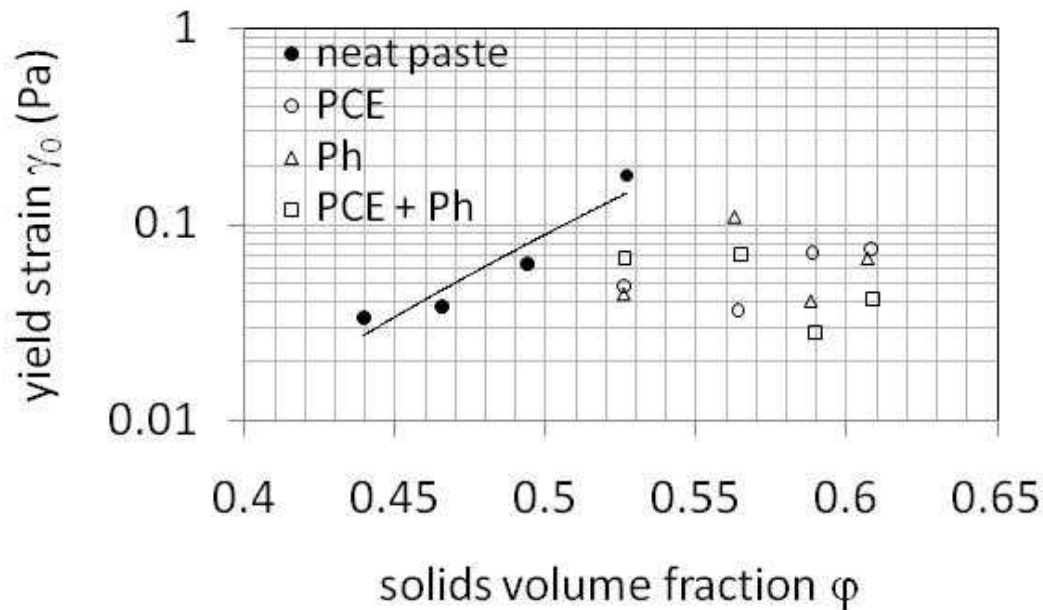
admixture dosage: PCE 3,66 kg/m³ of a 24% solution
 Ph 6,41 kg/m³ of a 30% solution
 PCE+Ph 5,04 kg/m³ of a 28% solution

blend PCE + Ph: 36% PCE + 64% Ph as dry matter

rheology of the mortar



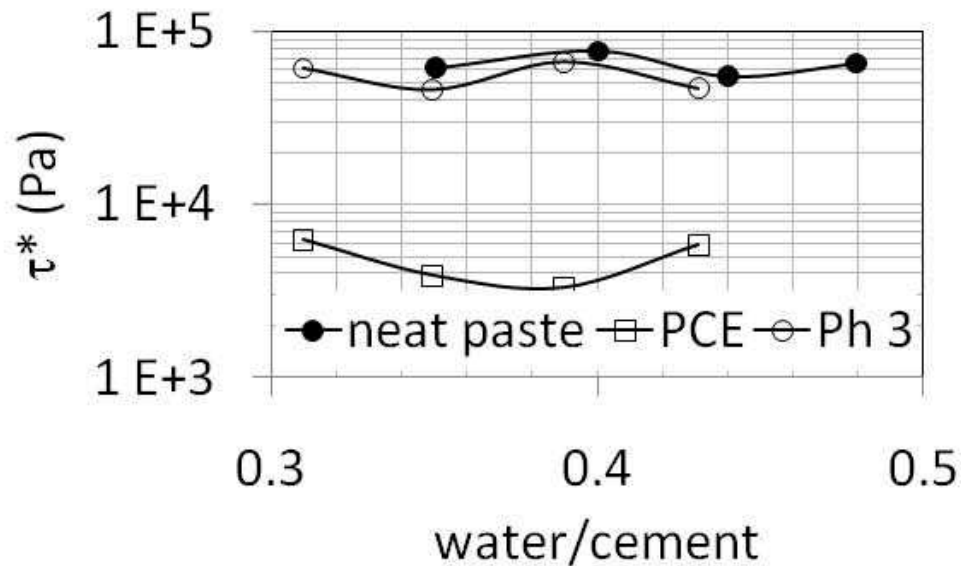
rheology of the paste



rheology of the paste

paste	neat	PCE	Ph	PCE + Ph
yield stress scaling exponent	9,98	39,1	20,0	23,6
yield strain scaling exponent	9,21	≈ 0	≈ 0	≈ 0

suggesting a different paste microstructure and inter-particle strength



concluding remarks

- cement paste rheology is controlled entirely by the attractive particle interactions in the dormant period
- the presence of superplasticizers modifies rheology but also microstructure – heterogeneity distribution

thank you!